ESTIMATION OF THE BORON DIFFUSION COEFFICIENTS IN FeB AND Fe₂B LAYERS DURING THE PACK-BORIDING OF A HIGH-ALLOY STEEL

DOLOČANJE KOEFICIENTA DIFUZIJE BORA V PLASTEH FeB IN Fe2B MED BORIRANJEM VISOKO LEGIRANEGA JEKLA V SKRINJI

Zahra Nait Abdellah^{1,2}, Mourad Keddam¹

¹Laboratoire de Technologie des Matériaux, Département de Sciences des Matériaux, Faculté de Génie Mécanique et Génie des Procédés, USTHB, B.P N°32, 16111 El-Alia, Bab-Ezzouar, Alger, Algérie ²Département de Chimie, Faculté des sciences, Université Mouloud Mammeri, 15000 Tizi-Ouzou, Algérie

keddam@yahoo.fr

Prejem rokopisa – received: 2013-03-31; sprejem za objavo – accepted for publication: 2013-06-07

In this work we propose a diffusion model to estimate the boron diffusion coefficients in FeB and Fe₂B layers during the pack-boriding of AISI M2 steel in the temperature range 1173–1323 K for a treatment time of 4–8 h. The proposed model is based on the mass-balance equations at the two growth fronts – FeB/Fe₂B and Fe₂B/substrate – under certain assumptions. The estimated values of the boron activation energies in the FeB and Fe₂B layers were compared with the literature data. The present model was extended to predict the thickness of each boride layer for the borided samples at different temperatures for 10 h. Iso-thickness diagrams were established to be used as a tool for predicting the thickness of each boride layer as a function of the two parameters: temperature and time. Finally, a simple equation was proposed to estimate the required time to obtain a single Fe₂B layer by diffusion annealing.

Keywords: boriding, incubation times, Fick's laws, simulation, growth kinetics, annealing

Predstavljeno delo predlaga model difuzije za določanje koeficienta difuzije bora v plasteh FeB in Fe₂B med boriranjem v skrinji jekla AISI M2 v temperaturnem območju 1173–1323 K pri spreminjanju trajanja postopka od 4 h do 8 h. Predlagani model temelji na enačbi masne balance na dveh rastočih mejnih ploskvah (FeB/Fe₂B) in (Fe₂B/osnova) pri določenih predpostavkah. Določena vrednost aktivacijske energije bora v FeB- in Fe₂B-plasti je bila primerjana s podatki iz literature. Predstavljeni model je bil razširjen, da bi lahko napovedal debelino vsake od obeh boridnih plasti za borirane vzorce pri različnih temperaturah in trajanju do 10 h. Postavljeni so bili diagrami enake debeline, ki so uporabni kot orodje za napovedovanje debeline vsakega od boriranih slojev v odvisnosti od dveh parametrov (temperature in časa). Predlagana je preprosta enačba za določanje potrebnega časa za nastanek plasti Fe₂B z difuzijskim žarjenjem.

Ključne besede: boriranje, inkubacijski čas, Fickovi zakoni, simulacija, kinetika rasti, žarjenje

1 INTRODUCTION

One of the surface-modification methods for improving the surface properties of ferrous and non-ferrous alloys is boriding. According to the Fe-B binary system, two kinds of iron borides, i.e., FeB and Fe₂B, with a narrow range of composition can be identified.¹ The boriding process applies in the temperature range 1073-1323 K between 1 h to 10 h and it can be carried out in solid, liquid or gaseous media. The possible formation of the FeB and Fe₂B iron borides depends upon various factors, such as the boron activity of the boriding medium, the chemical composition of the substrate, the process temperature and the treatment time. The morphology of the boride layers is influenced by the presence of alloying elements in the matrix. Saw-toothshaped layers are obtained in low-alloy steels, whereas in high-alloy steels, the interfaces tend to be flat. The modelling of the boriding kinetics is considered as a suitable tool to match the case depth with the intended industrial applications for this borided steel. So, the modelling of the growth kinetics for boride layers has gained much attention to simulate the boriding kinetics during recent decades. $^{\rm 2-26}$

In the present work an original diffusion model is proposed to estimate the boron diffusion coefficients in the FeB and Fe₂B layers grown on AISI M2 steel by considering the boride incubation times. A non-linear boron-concentration profile is assumed through the boride layers. The mass-balance equations were applied to the two diffusion fronts: the FeB/Fe₂B and Fe₂B/substrate interfaces in the temperature range 1173–1323 K. In addition, a simple equation was proposed to estimate the required time to obtain a single Fe₂B layer by diffusion annealing.

2 THE DIFFUSION MODEL

The model takes into account the FeB/Fe_2B bilayer growth on the saturated substrate with boron atoms, as shown in **Figure 1**.

 $C_{\rm up}^{\rm FeB}$ and $C_{\rm low}^{\rm FeB}$ (= 16.23 % B) are the upper and lower boron mass concentrations in the FeB, while $C_{\rm up}^{\rm FeB}$ (= 9 % B) and $C_{\rm low}^{\rm FeB}$ (= 8.83 % B) are, respectively, the upper



Distance from the surface (µm)

Figure 1: Boron concentration profile through the FeB/Fe₂B bilayer **Slika 1:** Profil koncentracije bora skozi plasti (FeB/Fe₂B)

and lower boron concentrations in the Fe₂B. C_{ads} denotes the adsorbed concentration of boron,¹⁵ while *u* is the position of the Fe₂B/Fe₂B interface, and *v* is the position of the Fe₂B/substrate interface. C_0 is the boron solubility in the matrix and is equal to $35 \cdot 10^{-4}$ % B.² The upper boron content in the FeB phase (C_{up}^{FeB}), imposed by the boriding medium, gives rise to the two iron borides: FeB and Fe₂B. From a thermodynamic point of view, the FeB phase exhibits a narrow composition range (of about the mole fraction x = 1 % B or the mass fraction w = 0.2 % B), as identified by Massalski.²⁷ The upper boron content in the FeB phase was taken in the composition range of mass fractions 16.25–16.43 % B to obtain a bilayer configuration consisting of the two iron borides, FeB and Fe₂B.

The following assumptions are considered during the formulation of the diffusion model:

- The kinetics is dominated by the diffusion-controlled mechanism
- The growth of the boride layers is a consequence of the boron diffusion perpendicular to the sample surface
- The range of homogeneity of the iron borides is about *x* = 1 % B
- The iron borides nucleate after a certain incubation time
- The boride layer is thin in comparison to the sample thickness
- A local equilibrium occurs at the phase interfaces
- A planar morphology is assumed for the phase interfaces
- The volume change during the phase transformation is ignored
- The diffusion coefficient of boron in each iron boride does not vary with the boron concentration and follows an Arrhenius relationship
- A uniform temperature is assumed throughout the sample
- The alloying elements have no effect on the boron diffusion

• The presence of porosity is neglected during the boron diffusion.

The initial conditions of the diffusion problem are set up as follows:

$$C_{\text{FeB}} \{ x(t \succ 0) = 0 \} = 0$$

$$C_{\text{Fe}_{2B}} \{ x(t \succ 0) = 0 \} = 0$$
(1)

$$C_{\text{Fe}} \{ x(t \succ 0) = 0 \} = 0$$

The boundary conditions are given by the following equations:

$$C_{\text{FeB}}\left\{x[t=t_0^{\text{FeB}}(T)]=0\right\} = C_{\text{up}}^{\text{FeB}} \text{ for } C_{\text{ads}} \succ 16.23wt.\%B(2)$$
$$C_{\text{FeB}}\left\{x[t=t_0^{\text{FeB}}(T)]=0\right\} = C_{\text{Lum}}^{\text{FeB}} \text{ for } C_{\text{ads}} \prec 16.23wt.\%B$$

 $C_{\text{Fe}_{2}B}\left\{x[t=t_{0}^{\text{Fe}_{2}B}(T)]=0\right\}=C_{\text{up}}^{\text{Fe}_{2}B}$ for

и

8.83*wt*.%B \prec C_{ads} \prec 16.23*wt*.%B and without the FeB phase: (4)

$$C_{\text{Fe}_{2}\text{B}}\left\{x[t=t_{0}^{\text{Fe}_{2}\text{B}}(T)]=0\right\}=C_{\text{low}}^{\text{Fe}_{2}\text{B}} \text{ for } C_{\text{ads}} \prec 8.83wt.\%B$$

and without the FeB phase: (5)

$$C_{\text{FeB}}(x(t=t)=u) = C_{\text{low}}^{\text{FeB}}$$
(6)

$$C_{\text{Fe}_{2}B}(x(t=t)=u) = C_{\text{up}}^{\text{Fe}_{2}B}$$
 (7)

$$C_{\text{Fe},B}(x(t=t)=v) = C_{\text{low}}^{\text{Fe},B}$$
(8)

$$C_{\rm Fe}(x(t=t)=v) = C_0$$
 (9)

The mass-balance equations²⁸ are given by the equations (10) and (11):

$$Y_{\text{FeB}}\left(\frac{\mathrm{d}u}{\mathrm{d}t}\right) = \left[J_{\text{B}}^{\text{FeB}} - J_{\text{B}}^{\text{Fe}_{2}\text{B}}\right]_{x=u}$$
(10)

$$v_{\mathrm{Fe}_{2}\mathrm{B}}\left(\frac{\mathrm{d}v}{\mathrm{d}t}\right) + w'\left(\frac{\mathrm{d}u}{\mathrm{d}t}\right) = \left[J_{\mathrm{B}}^{\mathrm{Fe}_{2}\mathrm{B}}\right]_{x=v}$$
(11)

with

$$w_{\text{FeB}} = \left[05 \times (C_{\text{up}}^{\text{FeB}} - C_{\text{low}}^{\text{FeB}}) + (C_{\text{low}}^{\text{FeB}} - C_{\text{up}}^{\text{Fe}_2B}) \right]$$
$$w_{\text{Fe}_2B} = \left[05 \times (C_{\text{up}}^{\text{Fe}_2B} - C_{\text{low}}^{\text{Fe}_2B}) + (C_{\text{low}}^{\text{Fe}_2B} - C_{0}) \right]$$
$$w' = 05 \times (C_{\text{up}}^{\text{Fe}_2B} - C_{\text{low}}^{\text{Fe}_2B})$$

The boron flux through a given boride layer is obtained from Fick's first law as follows:

$$J_{\rm B}^{i} = -D_{\rm B}^{i} \frac{\partial C_{i}(x,t)}{\partial x} \text{ with } i = (\text{FeB or Fe}_{2}\text{B})$$
(12)

 $D_{\rm B}^{\rm FeB}$ and $D_{\rm B}^{\rm Fe_2B}$ are, respectively, the diffusion coefficients of boron in the FeB and Fe₂B phases. The boron concentration profile in the FeB layer is given by:

$$C_{\text{FeB}}(x,t) = C_{\text{up}}^{\text{FeB}} + \frac{(C_{\text{low}}^{\text{FeB}} - C_{\text{up}}^{\text{FeB}})}{erf\left(\frac{u}{2\sqrt{D_{\text{B}}^{\text{FeB}}t}}\right)} \cdot erf\left(\frac{x}{2\sqrt{D_{\text{B}}^{\text{FeB}}t}}\right)$$

For $0 \le x \le u$

(13)

In the same way, the boron concentration profile in the Fe_2B layer can be obtained as follows:

Materiali in tehnologije / Materials and technology 48 (2014) 2, 237-242

$$C_{\text{Fe}_{2}\text{B}}(x,t) = C_{\text{up}}^{\text{Fe}_{2}\text{B}} + \frac{(C_{\text{low}}^{\text{Fe}_{2}\text{B}} - C_{\text{up}}^{\text{Fe}_{2}\text{B}})}{\left[erf\left(\frac{u}{2\sqrt{D_{\text{B}}^{\text{Fe}_{2}\text{B}}t}}\right) - erf\left(\frac{v}{2\sqrt{D_{\text{B}}^{\text{Fe}_{2}\text{B}}t}}\right) \right]} \right]$$
$$\left[erf\left(\frac{u}{2\sqrt{D_{\text{B}}^{\text{Fe}_{2}\text{B}}t}}\right) - erf\left(\frac{x}{2\sqrt{D_{\text{B}}^{\text{Fe}_{2}\text{B}}t}}\right) \right]$$
For $u \le x \le v$ (14)

For $u \le x \le v$

The FeB layer thickness u grows parabolically according to equation (15), where k_{FeB} represents the parabolic growth constant at the FeB/Fe₂B interface:

$$u = k_{\rm FeB} \left[t - t_0^{\rm FeB}(T) \right]^{1/2}$$
(15)

The distance v is the location of the Fe_2B /substrate interface and k its parabolic growth constant (equation (16)) and the difference (l = v - u) denotes the layer thickness of the Fe₂B (equation 17):

$$=k[t-t_{0}(T)]^{1/2}$$
(16)

$$l = v - u = k \left[t - t_0(T) \right]^{1/2} - k_{\text{FeB}} \left[t - t_0^{\text{FeB}}(T) \right]^{1/2}$$
(17)

with $t_0^{\text{FeB}}(T) \succ t_0(T)$ and $k \succ k_{\text{FeB}}$

where $t_0(T)$ is the boride incubation time of the total boride layer and $t_0^{\text{FeB}}(T)$ is the boride incubation time of the FeB layer. To take into account the effect of the boride incubation times when solving the mass-balance equations, it is necessary to define the two parameters $\beta_{\text{FeB}}(T)$ and $\beta(T)$:

$$\beta_{\rm FeB}(T) = \left[1 - \frac{t_0^{\rm FeB}(T)}{t}\right]^{0.5}$$
(18)

and

The layer thickness of the FeB (u) is related to the $\beta_{\text{FeB}}(T)$ parameter by equation (20):

 $\beta(T) = \left[1 - \frac{t_0(T)}{t}\right]^{0.5}$

$$u = k_{\rm FeB} \beta_{\rm FeB}(T) \sqrt{t} \tag{20}$$

(19)

and

In the same way, the layer thickness of the $Fe_2B(l)$ is expressed using equation (21):

$$l = \left[k\beta(T) - k_{\text{FeB}}\beta_{\text{FeB}}(T)\right]\sqrt{t}$$
(21)

3 ESTIMATION OF THE BORON DIFFUSION COEFFICIENTS IN THE FeB AND Fe2B LAYERS

To estimate the boron diffusion coefficients in the FeB and Fe₂B layers, the experimental results published by Campos-Silva et al.29 on borided AISI M2 steel were used. In this reference work, the powder-pack boriding was carried out at four temperatures, (1173, 1223, 1273 and 1323) K, for three exposure times, (4, 6 and 8) h, using the B₄C Durborid as a boriding medium. Eighty measurements were performed on different cross-sections of the borided samples from the AISI M2 steel to determine the thickness of each boride layer.

Tables 1 and 2 list the experimental parabolic growth constants for each phase interface with the corresponding incubation times. The experimental values of the parabolic growth constants at each phase interface were obtained from the slopes of the curves relating the squared boride layer thickness to the boriding time. The boride incubation times were deduced for a null boride layer thickness.

Table 1: Experimental values of the parabolic growth constants at the FeB/Fe2B interface in the temperature range 1173-1323 K with the corresponding boride incubation times

Tabela 1: Eksperimentalne vrednosti konstant parabolične rasti na stiku (FeB/Fe2B) v temperaturnem območju 1173-1323 K, z ustreznim inkubacijskim časom borida

<i>T</i> /K	Experimental growth Constants: $k_{\text{FeB}}/(\mu \text{m s}^{-1/2})$	$t_0^{\text{FeB}}(T)/\text{s}$
1173	0.065	10131
1223	0.121	6085.7
1273	0.179	4347.8
1323	0.238	3815.5

Table 2: Experimental values of the parabolic growth constants at the Fe₂B/substrate interface in the temperature range 1173-1323 K with the corresponding boride incubation times

Tabela 2: Eksperimentalne vrednosti konstant parabolične rasti na stiku (Fe₂B/podlaga) v temperaturnem območju 1173-1323 K, z ustreznim inkubacijskim časom borida

<i>T</i> /K	Experimental growth Constants: $k/(\mu m s^{-1/2})$	$t_0(T)/s$
1173	0.168	8806.2
1223	0.305 0.448	4323
1323	0.589	3742.7

It was demonstrated that the higher boriding temperatures involve the shorter incubation times,²⁴ as shown in Tables 1 and 2. The two respective parameters $\beta_{\text{FeB}}(T)$ and $\beta(T)$ are linearly dependent on the boriding temperature and can be approximated by equations (22) and (23) from a linear fitting of the experimental data displayed in Figure 2:

$$\beta_{\rm FeB}(T) = (1.39 \times 10^{-3} T - 0.8579)$$
(22)

 $\beta(T) = (1 \cdot 40 \times 10^{-3} T - 0.8470)$

For this purpose, a computer code written in Matlab (version 6.5) was used to estimate the boron diffusivity in each boride layer. This program requires the following input data: the time, the temperature, the lower and upper boron concentrations at each phase interface as well as the two parameters $\beta_{\text{FeB}}(T)$ and $\beta(T)$. By solving the mass-balance equations (equations (10) and (11)) via the Newton-Raphson method,³⁰ it is possible to determine the boron diffusion coefficients in the FeB and Fe2B layers. Table 3 summarizes the estimated values of the boron diffusion coefficients in the FeB and Fe₂B layers for an upper boron content equal to w = 16.40 % in the FeB phase.

Figure 3 depicts the temperature dependence of the boron diffusion coefficients in the FeB and Fe₂B layers according to the Arrhenius equation. The value of the

(23)



Figure 2: Evolution of the two parameters as a function of the boriding temperature: a) $\beta_{\text{FeB}}(T)$ and b) $\beta(T)$

Slika 2: Razvoj dveh parametrov v odvisnosti od temperature boriranja: a) $\beta_{FeB}(T)$ in b) $\beta(T)$

Table 3: Determination of the boron diffusion coefficient in each boride layer for an upper boron mass fraction content of w = 16.40 % in the FeB layer

Tabela 3: Določanje koeficienta difuzije bora v vsaki boridni plasti za zgornji masni delež vsebnosti bora 16,40 % v plasti FeB

T/K	$D_{\rm B}^{\rm FeB}({\rm m}^2{\rm s}^{-1})\times 10^{-12}$	$D_{\rm B}^{{\rm Fe_2B}}({\rm m^2s^{-1}}) \times 10^{-12}$
1173	0.376	0.462
1223	1.282	1.502
1273	2.797	3.227
1323	4.915	5.539

boron activation energy in each boride layer can be easily obtained from the slopes of the corresponding curves. So, the boron diffusion coefficients in the FeB and Fe₂B layers are, respectively, given by equations (24) and (25):

$$D_{\rm B}^{\rm FeB} = 2.8 \times 10^{-3} \exp^{\frac{-220.2 \, \text{kJ/mol}}{RT}} (\text{m}^2 \,\text{s}^{-1})$$
(24)



Figure 3: An Arrhenius relationship between the boron diffusion coefficient and the temperature: a) FeB layer, b) Fe₂B layer **Slika 3:** Arrheniusova odvisnost med koeficientom difuzije bora in temperaturo: a) FeB-plast, b) Fe₂B-plast

$$D_{\rm B}^{\rm Fe_2B} = 1.6 \times 10^{-3} \exp \frac{-213 \, \text{kJ/mol}}{RT} (\text{m}^2 \, \text{s}^{-1})$$
 (25)

where *R* is the universal gas constant (= 8.314 J/(mol K)), and *T* represents the absolute temperature in Kelvin.

The reported values of the activation energies^{24,29,31-33} of the borided steels are listed in **Table 4** together with the values from this work. The obtained values of the activation energies are found to be dependent on the boriding method and on the chemical composition of the substrates.

In **Table 5**, a comparison was achieved between the experimental boride layer thicknesses and the simulated ones at different temperatures for 10 h of treatment. The simulated results were obtained from equations (20) and (21). The present model was able to predict the boride layer thickness (FeB or Fe_2B) for the given boriding conditions.

 Table 4: Values of the boron activation energies obtained for different borided steels

 Tabela 4: Vrednosti aktivacijske energije bora, dobljene iz različnih boriranih jekel

Material	Boriding method	Activation energy of FeB E/(kJ mol ⁻¹)	Activation energy of Fe_2B $E/(kJ mol^{-1})$	Reference
AISI M2	Paste	283	239.4	32
AISI4140	Paste	_	168.5	33
AISI H13	Powder-pack	_	186.2	31
AISI 316L	Powder-pack	204	198	24
AISI M2	Powder-pack	223	207	29
AISI M2	Powder-pack	220.2	213	Present study

Table 5: Experimental (exp.) and simulated (sim.) values of the boride layer thickness in the temperature range 1173–1323 K for 10 h of treatment, with an upper boron mass fraction of content of w = 16.40 % in the FeB phase

Tabela 5: Eksperimentalne (exp.) in simulirane (sim.) vrednosti za debelino plasti borida v temperaturnem območju 1173–1323 K, za 10 h obdelave pri gornjem masnem deležu vsebnosti bora 16,40 %, v FeB plasti

<i>T</i> /K	FeB (µm) exp.	FeB (µm) sim.	Fe ₂ B (µm) exp.	Fe ₂ B (µm) sim.
1173	10.17	10.30	19.66	16.70
1223	20.98	17.89	32.81	28.23
1273	28.30	29.75	51.83	45.81
1323	40.24	47.60	72.28	71.67

In **Table 6**, the predicted values of the boride layer thicknesses are compared with the experimentally determined values in the temperature range 1173-1323 K for a treatment time varying from 4 h to 8 h. Good agreement was observed between the experimental data and the simulation results for an upper boron content equal to w = 16.40 % in the FeB phase.

Table 6: Experimental (exp.) and simulated values (sim.) of the boride layer thickness in the temperature range 1173-1323 K for different treatment times with an upper boron content w = 16.40 % in the FeB phase

Tabela 6: Eksperimentalne (exp.) in simulirane (sim.) vrednosti debeline plasti borida pri temperaturah 1173–1323 K za različne čase obdelave in zgornjo vsebnostjo bora w = 16,40 % v FeB-fazi

T/K Time		FeB (µm)	FeB (µm)	Fe ₂ B (µm)	Fe ₂ B (µm)
1/1	(h)	exp.	sim.	exp.	sim.
	4	4.24	6.47	8.31	10.53
1173	6	6.96	7.92	12.04	12.89
	8	8.88	9.15	14.87	14.89
	4	11.03	11.32	18.96	17.85
1223	6	15.07	13.86	24.54	21.86
	8	18.23	16.00	29.08	25.24
1273	4	17.94	18.81	27.02	28.97
	6	23.51	23.04	35.37	35.48
	8	28.00	26.60	42.09	40.97
1323	4	24.48	24.89	36.31	35.90
	6	31.73	32.27	46.97	46.43
	8	37.61	38.25	55.61	54.98

Figure 4 displays the iso-thickness diagrams describing the evolution of the boride layer thickness as a function of the time and the boriding temperature. The results derived from **Figure 4** can be used as a tool to predict the boride layer thickness in relation with its practical use in an industrial area.

4 OBTAINING OF A SINGLE LAYER OF Fe₂B BY DIFFUSION ANNEALING

In industrial practice it is possible to reduce the brittleness of boride layers by controlling their microstructure. It is known that a single Fe₂B boride layer is more desirable than a dual FeB-Fe₂B layer.³⁴ This makes it possible to reduce the FeB layer thickness by applying a diffusion annealing in a hydrogen atmosphere. During this stage, the supply of boron is stopped since the concentration gradient of boron in the FeB is null (i.e., $C_{up}^{\text{FeB}} = C_{low}^{\text{FeB}} = 16.23 \%$), the FeB layer will be converted into an Fe₂B layer. The time required to eliminate the FeB layer during the diffusion annealing can be obtained from equation (26):

$$t_{u_{FeB}=0} = \frac{u \times l \times (C_{low}^{FeB} - C_{up}^{Fe_2B})}{D_B^{Fe_2B} (C_{up}^{Fe_2B} - C_{low}^{Fe_2B})}$$
(26)

where *u* is the FeB layer thickness (μ m), *l* the Fe₂B layer thickness (μ m) and $D_{B}^{Fe_{2}B}$ represents the boron diffusion coefficient in Fe₂B. It is clear that the annealing time depends on the boron diffusion coefficient in Fe₂B, and also on the thickness of each boride layer. During the diffusion annealing, an infinitesimal reduction of the FeB layer is related to the infinitesimal growth of the Fe₂B layer by equation (27):

$$\Delta u = -\left(\frac{w_{\text{Fe}_2\text{B}}}{w_{\text{Fe}\text{B}} + w_{\text{Fe}_2\text{B}} + w'}\right) \Delta l = -0.5493 \Delta l \qquad (27)$$

The value of the Fe₂B layer thickness l' (µm) after diffusion annealing becomes:

$$l' = l + \frac{\Delta u}{0.5493} \tag{28}$$

Table 7 presents the simulation results obtained from equations (26) and (28) to estimate the Fe_2B layer thickness after diffusion annealing and the time required



Figure 4: Iso-thickness diagrams describing the evolution of the boride layers: a) FeB, b) Fe_2B

Slika 4: Diagram enakih debelin opisuje razvoj boridnih plasti: a) FeB, b) Fe2B

Z. NAIT ABDELLAH, M. KEDDAM: ESTIMATION OF THE BORON DIFFUSION COEFFICIENTS IN FeB AND Fe2B LAYERS ...

to eliminate the FeB layer in the case of the borided samples treated at different temperatures for 10 h. The obtained annealing times are increased with an increase of the boriding temperature since the boride layer becomes thicker. In this context, Kulka et al.²⁶ have experimentally determined the annealing time using a hydrogen atmosphere to obtain a single Fe₂B layer on gas borided Armco Fe at 1173 K for 2 h in a gas mixture (H₂–BCl₃). They found that the total elimination of the FeB layer took about 1 h.

Furthermore, it was shown by Dybkov et al.³⁵ that annealing of a borided Fe–Cr sample for 6 h resulted in the disappearance of the FeB layer.

Table 7: Estimation of the Fe₂B layer thickness and the time required to eliminate the FeB layer for the borided samples at different temperatures for 10 h

<i>T/</i> K	FeB (µm) sim.	Fe ₂ B (µm) sim.	Fe ₂ B (µm) After diffusion annealing Equation (28)	Annealing time $t_{u_{\text{FeB}}=0}/h$ Equation (26)
1173	10.30	16.70	35.45	4.15
1223	17.89	28.23	60.79	4.99
1273	29.75	45.81	99.96	5.92
1323	47.60	71.67	158.32	6.93

Tabela 7: Določanje debeline plasti Fe₂B in čas, potreben za odpravo FeB plasti, za vzorce, borirane 10 h pri različnih temperaturah

5 CONCLUSIONS

In this work an original diffusion model was proposed to estimate the boron diffusion coefficients in the FeB and Fe₂B layers grown on AISI M2 steel. To determine the boron activation energy in each boride layer, the mass-balance equations were formulated, including the effect of the boride incubation times. The estimated boron activation energies were compared with the literature data. The present model was extended to predict the thickness of each boride layer for the borided samples at different temperatures for 10 h. Iso-thickness diagrams were established to be used as a tool to predict the thickness of each boride layer as a function of the two parameters (temperature and time). The required time to obtain a single Fe₂B layer by diffusion annealing was estimated on the basis of a simple equation. The formation of a single Fe₂B layer on AISI M2 steel depended on the boriding parameters.

Acknowledgements

This work was carried out in the framework of the CNEPRU project under code number J0300220100093 of the Algerian Ministry of Higher Education and Scientific Research.

6 REFERENCES

¹S. A. Sinha, Boriding, J. Heat Treat., 4 (1991), 437–447
 ²M. Keddam, Appl. Surf. Sci., 253 (2006), 757–761

- ⁴ M. Keddam, S. M. Chentouf, Appl. Surf. Sci., 252 (2005), 393–399
- ⁵ R. D. Ramdan, T. Takaki, Y. Tomita, Mater. Trans., 49 (**2008**), 2625–2631
- ⁶ M. Keddam, Defect Diffus. Forum, 273–276 (**2008**), 318–322
- ⁷I. Campos, M. Islas, G. Ramírez, L. Zuniga, C. Villa Velázquez, C. Mota, Appl. Surf. Sci., 253 (2007), 6226–6231
- ⁸ M. Keddam, Appl. Surf. Sci., 236 (2004), 451–455
- ⁹ I. Campos, G. Ramírez, U. Figueroa, C. V. Velázquez, Surf. Eng., 23 (2007), 216–222
- ¹⁰ I. Campos, G. Ramírez, U. Figueroa, J. Martinez, O. Morales, Appl. Surf. Sci., 253 (**2007**), 3469–3475
- ¹¹ D. S. Kukharev, S. P. Fizenko, S. I. Shabunya, J. Eng. Phys. Therm., 69 (**1996**), 187–193
- ¹² C. M. Brakman, A. J. W. Gommers, E. J. Mittemeijer, J. Mater. Res., 4 (**1989**), 1354–1370
- ¹³ I. Campos-Silva, M. Ortíz-Domínguez, Int. J. Microstructure and Materials Properties, 5 (2010), 26–38
- ¹⁴ M. Ortiz-Domínguez, E. Hernandez-Sanchez, J. Martinez-Trinídad, M. Keddam, I. Campos-Silva, Kovove Materialy, 48 (2010), 1–6
- ¹⁵ L. G. Yu, X. J. Chen, K. A. Khor, G. Sundararajan, Acta Mater., 53 (2005), 2361–2368
- ¹⁶ M. Keddam, R. Chegroune, Appl. Surf. Sci., 256 (**2010**), 5025–5030
- ¹⁷ M. Keddam, M. Ortiz-Domínguez, I. Campos-Silva, J. Martinez-Trinídad, J. Appl. Surf. Sci., 256 (2010), 3128–3132
- ¹⁸ M. Keddam, Defect Diffus. Forum, 297–301 (2010), 269–274
- ¹⁹ M. Ortiz-Domínguez, I. Campos-Silva, J. Martínez-Trinidad, M. Elías-Espinosa, E. Hernández-Sánchez, D. Bravo-Bárcenas, Defect Diffus. Forum, 297–301 (2010), 294–299
- ²⁰ I. Campos-Silva, M. Ortiz-Domínguez, M. Keddam, N. López-Perrusquia, A. Carmona-Vargas, M. Elias-Espinosa, Appl. Surf. Sci., 255 (**2009**), 9290–9295
- ²¹I. Campos-Silva, N. López-Perrusquia, M. Ortiz-Domínguez, U. Figueroa-López, O. A. Gómes-Vargas, A. Meneses-Amador, G. Rodríguez-Castro, Kovove Mater., 47 (2009), 1–7
- ²² I. Campos-Silva, M. Ortiz-Domínguez, C. Villa Velazquez, R. Escobar, N. López, Defect Diffus. Forum, 272 (2007), 79–86
- ²³ I. Campos-Silva, M. Ortiz-Domínguez, N. López-Perrusquia, R. Escobar-Galindo, O. A. Gómez-Vargas, E. Hernández-Sánchez, Defect Diffus. Forum, 283–286 (2008), 681–686
- ²⁴ I. Campos-Silva, M. Ortiz-Domínguez, O. Bravo-Bárcenas, M. A. Doñu-Ruiz, D. Bravo-Bárcenas, C. Tapia-Quintero, M. Y. Jiménez-Reyes, Surface and Coatings Technology, 205 (**2010**), 403–412
- ²⁵ M. Keddam, Appl. Surf. Sci., 257 (2011), 2004–2010
- ²⁶ M. Kulka, N. Makuch, A. Pertek, L. Maldzinski, Journal of Solid State Chemistry, 199 (2013), 196–203
- ²⁷ T. B. Massalski, Binary Alloys Phase diagrams, 2nd ed., ASM International, Metals Park, OH 1990
- ²⁸ M. Keddam, M. E. Djeghlal, L. Barrallier, Mater. Sci. Eng., A 378 (2004), 475–478
- ²⁹ I. Campos-Silva, M. Ortiz-Domínguez, C. Tapia-Quintero, G. Rodriguez-Castro, M. Y. Jiménez-Reyes, E. Chavez-Gutiérrez, Journal of Materials Engineering and Performance, 21 (2012) 8, 1714–1723
- ³⁰ W. H. Press, B. P. Flannery, S. A. Teukolsky, Numerical recipes in Pascal: the art of scientific computing, Cambridge University, 1989
- ³¹ K. Genel, Vacuum, 80 (2006), 451–457
- ³² I. Campos, R. Torres, O. Bautista, G. Ramirez, L. Zuniga, Appl. Surf. Sci., 252 (2006), 2396–2403
- ³³Z. Nait Abdellah, M. Keddam, A. Elias, International Journal of Materials Research, 104 (2013), 260–265
- ³⁴ G. Celebi, M. Ipek, C. Bindal, A. H. Ucisik, Materials Forum, 29 (2005), 456–460
- ³⁵ V. I. Dybkov, W. Lengauer, K. Barmak, Journal of Alloys and Compounds, 398 (2005), 113–122

³ M. Keddam, Int. J. Mater. Res., 100 (**2009**), 901–905